

# Tracking the insulator-to-metal phase transition in VO<sub>2</sub> with few-femtosecond extreme UV transient absorption spectroscopy

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Coulomb correlations can manifest in exotic properties in solids, but how these properties can be accessed and ultimately manipulated in real time is not well understood. The insulator-to-metal phase transition in vanadium dioxide (VO<sub>2</sub>) is a canonical example of such correlations. Here, few-femtosecond extreme UV transient absorption spectroscopy (FXTAS) at the vanadium  $M_{2,3}$  edge is used to track the insulator-to-metal phase transition in VO<sub>2</sub>. This technique allows observation of the bulk material in real time, follows the photoexcitation process in both the insulating and metallic phases, probes the subsequent relaxation in the metallic phase, and measures the phase-transition dynamics in the insulating phase. An understanding of the VO<sub>2</sub> absorption spectrum in the extreme UV is developed using atomic cluster model calculations, revealing  $V^{3+}/d^2$  character of the vanadium center. We find that the insulator-to-metal phase transition occurs on a timescale of  $26 \pm 6$  fs and leaves the system in a long-lived excited state of the metallic phase, driven by a change in orbital occupation. Potential interpretations based on electronic screening effects and lattice dynamics are discussed. A Mott-Hubbard-type mechanism is favored, as the observed timescales and  $d^2$  nature of the vanadium metal centers are inconsistent with a Peierls driving force. The findings provide a combined experimental and theoretical roadmap for using time-resolved extreme UV spectroscopy to investigate nonequilibrium dynamics in strongly correlated materials.

ultrafast dynamics | vanadium dioxide | insulator-to-metal transition | extreme UV

The Coulomb interaction of charges in a solid depends sensitively on local screening, bonding structure, and orbital occupancy. These electronic correlation effects are known to manifest themselves in unusual properties, such as superconductivity, colossal magnetoresistance, and insulator-to-metal phase transitions (IMTs), which can be switched on or off via small perturbations. Understanding if and how these correlation-driven properties can be manipulated in real time will open the door to using these materials as ultrafast photonic switches and will establish the electronic speed limits for next-generation devices (1–3).

Following the real-time dynamics of carrier interactions necessarily requires time-resolving the excitation and relaxation of electron correlations. Studies at longer timescales are complicated by simultaneous effects of structural distortion together with carrier screening and thermalization. In contrast, few-femtosecond extreme UV absorption spectroscopy (FXTAS) provides temporal resolution close to the Fourier limit for single-photon excitation with broadband visible light. It has the capability to separate electronic and structural effects on the basis of their intrinsic timescales and can isolate early-time electronic dynamics spectroscopically via atom-specific core-level electronic transitions (4, 5). Previous few-femtosecond and attosecond measurements of electron dynamics in solid-state systems have been restricted to simple band insulators or semiconductors and metals in which dynamics are initiated by the interaction with strong electric fields

(4, 6–11). We report here on FXTAS measurements of the IMT process following single-photon excitation in vanadium dioxide (VO<sub>2</sub>), a promising material for strongly correlated devices, and resolve the timescale of the IMT.

The insulating phase of VO<sub>2</sub> has monoclinic symmetry with dimerized pairs of vanadium atoms forming 1D chains within the crystal structure. As illustrated in Fig. 1, these bonds are broken and the crystal adopts a more symmetric rutile structure upon transition to the metallic phase at 340 K. The relative significance of the structural distortion in mediating the IMT has long been debated, and the driving force of the IMT in VO<sub>2</sub> has been described alternatively as the result of a Peierls-like structural instability, electron correlation and Mott-Hubbard effects, or a complex interplay of both (12–18). Previous time-resolved experiments on VO<sub>2</sub> have provided intriguing results: early optical (19, 20) and X-ray (21) measurements established subpicosecond switching times, measuring shorter and shorter timescales until an 80-fs bottleneck for the phase transition was observed in 2004 (22). However, more recent experiments in time-resolved photoemission (23) and transient reflectivity (24) have not observed the

## Significance

An insulator-to-metal phase transition is a process that changes a solid material from being electrically nonconductive to being conductive. The phase transition in vanadium dioxide is a well-studied example where the process can occur in less than a picosecond, making it exciting for ultrafast electronic switches. This paper measures a record speed for the phase transition of 26 fs into a long-lived excited state of the metal that persists out to >60 ps. The extreme UV absorption spectrum of the material is also measured and (together with the ultrafast timescale) belies a structural mechanism that has long been deliberated. The measured femtosecond timescale provides fundamental insight into the electronic speed limits of these complex phenomena.

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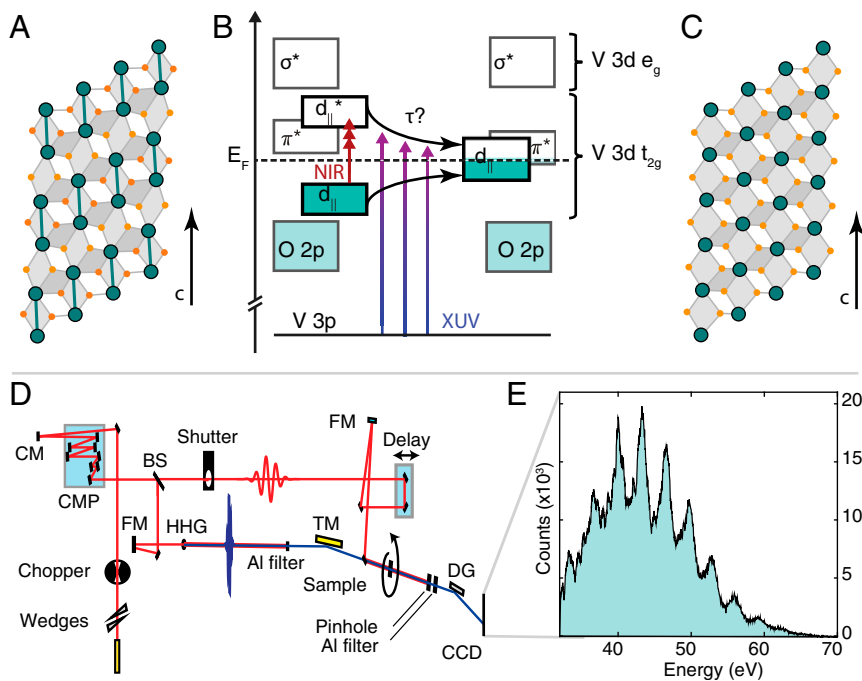
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**Fig. 1.** (A) Insulating ground state of VO<sub>2</sub> (V atoms: gray, O atoms: orange) exhibits a monoclinic crystal structure with dimerized chains of vanadium atoms along the *c* axis and two different types of oxygen sites, those between the bonded vanadium atoms, and those between the nonbonded vanadium atoms. (B) Electronic structure is depicted and can be understood via crystal field theory, where the *d* orbitals denoted *d*<sub>||</sub> align parallel to the *c* axis of the crystal. Electrons excited across the band gap with a few-femtosecond NIR pulse initiate the IMT, and the progress is tracked via core-level transitions into the unoccupied conduction and valence band states using time-delayed attosecond pulses. (C) Final metallic state of the system has a rutile crystal structure in which the vanadium dimerization has been melted. (D) Optical layout of the experiment: CM, collimating mirror; CMP, chirped mirror compressor; BS, beam splitter; FM, focusing mirror; HHG, high harmonic generation; DG, diffraction grating. (E) Representative high harmonic spectrum used in the experiment, at 100-ms exposure time and 100-Hz repetition rate.

bottleneck, and measured speeds <60 fs, which were limited by the temporal resolution of the experiment. In single-crystal microrods, timescales were found to vary from 200 fs (nonpulse duration limited) to 40 fs (pulse duration limited) (25). Time-resolved electron diffraction (26, 27) has established that dilation of the V–V dimers occurs within ~300 fs, and a coherent stretching of this mode has been observed following the phase transition (22, 24, 28), but the role of electronic versus structural effects at the earliest timescales is still not well understood. The work here establishes a  $26 \pm 6$ -fs timescale for the phase transition following single-photon excitation with a few-cycle femtosecond laser pulse, not limited by the  $5.6 \pm 0.8$ -fs instrument response function measured in situ. We note that the ultrafast timescale, coupled with the finding of a *d*<sup>2</sup> ground state, is inconsistent with a structural Peierls transition, and lends support for Mott transition mechanisms.

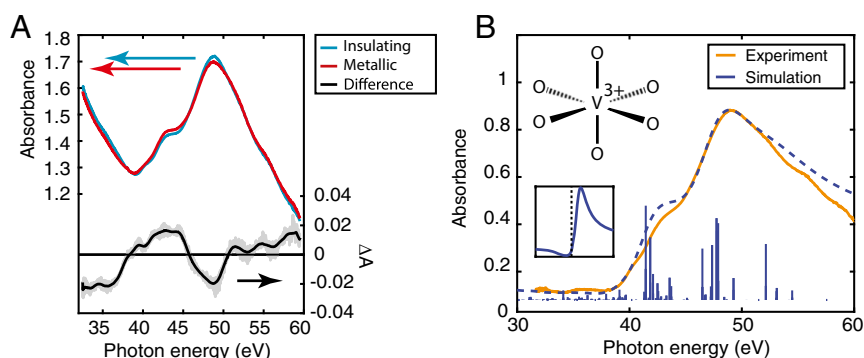
## Results

**Static Extreme UV Absorption Measurements.** The static extreme UV (XUV) *M*<sub>2,3</sub>-edge absorption spectrum of VO<sub>2</sub> is measured for both the insulating and heat-induced metallic phases in transmission geometry, using attosecond pulses that cover a spectral range from 30 to 60 eV. The samples are 25-nm-thick polycrystalline thin films of VO<sub>2</sub> deposited on 30-nm-thick freestanding Si<sub>3</sub>N<sub>4</sub> substrates, which are rapidly rotated to prevent the accumulation of heat from successive laser pulses (*Materials and Methods*). The transient evolution of this vanadium edge structure is followed in real time via FXTAS as schematically illustrated in Fig. 1. A sub-5-fs near-infrared (NIR) laser pulse centered at 760 nm impinges on VO<sub>2</sub> in its insulating phase, serving as the pump step to induce the IMT. Time-delayed broadband XUV attosecond pulses are then collinearly transmitted through the VO<sub>2</sub> thin film, probing the core-level absorption of the bulk material around the vanadium 3*p* transition (*M*<sub>2,3</sub> edge) at 39 eV (29). The instrument response function is monitored in situ via transient absorption of neon atoms and is measured at  $5.6 \pm 0.8$  fs (*Neon Transient Absorption Analysis for Time-Delay Calibration*). The time-dependent changes in absorption are measured directly via changes in the optical density ( $\Delta OD$ ) in a NIR-pump-on/NIR-pump-off measurement. Transient absorption experiments are carried out on both the low-temperature insulating phase and the higher-temperature metallic phase of VO<sub>2</sub> (produced

via a quartz halogen heat lamp) at an incident laser fluence of 25 mJ/cm<sup>2</sup>, well above the fluence threshold to induce the phase transition (see ref. 26 and data in *Fluence Dependence Measurements*). The experiments on the hot metallic phase serve as a reference for the ultrafast response to interband electronic excitations and provide a direct measurement of the XUV absorption spectrum of the photoexcited metallic phase.

The measured XUV static absorption spectrum around the vanadium *M*<sub>2,3</sub> edge is shown in Fig. 2*A*, for both the insulating and metallic phases. The signal at low energies (*E* < 39 eV) is assigned to transitions into the continuum from the oxygen 2*s* core level and from the oxygen-2*p*–vanadium-3*d* hybridized valence band. This ionization-type background is fit using the signal below 35 eV, well below the onset of the vanadium *M* edge, with a 1/*E*-type form, which provided the best fit between 20 and 35 eV based on the data in the CXRO database (30). This background is subtracted across all energies to obtain the experimental *M*<sub>2,3</sub>-edge spectrum shown in Fig. 2*B*.

The *M*<sub>2,3</sub>-edge absorption spectrum is simulated using atomic multiplet theory, as the *L*- and *M*-edge spectra of transition metals are known to be dominated by localized atomic effects (31) and cannot be treated solely as a partial density of states of the conduction band. While consideration of formal oxidation states might suggest modeling VO<sub>2</sub> as a V<sup>4+</sup>, or *d*<sup>1</sup>, metal center, there is considerable evidence that, due to strong hybridization, the vanadium site in VO<sub>2</sub> is actually closer to *d*<sup>2</sup> electron occupation (13, 32–34). Indeed, it is found that the experimental *M*<sub>2,3</sub>-edge spectrum is best matched with a simulated V<sup>3+</sup>, or *d*<sup>2</sup>, metal center. Further details of the calculations can be found in *Atomic Multiplet Simulations*. The discrete transition energies and amplitudes for the V<sup>3+</sup> system are shown in Fig. 2*B*. The transition lines are multiplied by a Gaussian function (to account for the experimental energy resolution) and convoluted with a Fano lineshape (to account for final-state interference effects and lifetime broadening), generating the full *M*-edge spectrum, as described in *Supporting Information*. A Fano parameter of *q* = 2.4 provides the best fit to the data, and lies well in the range of typical Fano parameters for transition-metal *M* edges [*q* = 0.7–3.5 (35–38)]. The spectral shape of the measured core-level transition is in excellent agreement with XUV reflectivity measurements of single crystals in the same energy range (39).

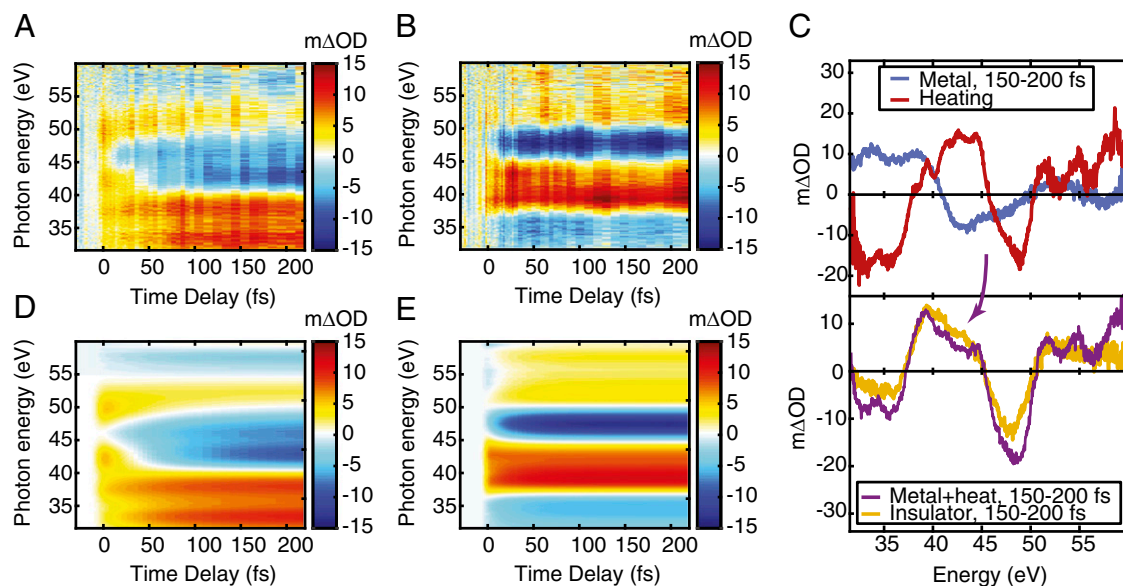


**Fig. 2.** (A) Static XUV absorbance spectra of the insulating (blue) and metallic (red) phases of  $\text{VO}_2$  are presented on the left-hand y axis. The difference in absorbance (black) is shown magnified in the lower part of the figure, on the right-hand y axis, with error bars shown in gray. (B) Vanadium  $M_{2,3}$  edge (orange) of the insulating phase spectrum after removing the low-energy ionization contribution. A cluster-model calculation (blue) is shown for a  $\text{V}^{3+}$  system. The transition energies and amplitudes are convoluted with the inset Fano profile to account for final-state interference effects, which maps the stick spectrum to the final simulated spectrum (blue dashed line), where due to the Fano asymmetry, spectral weight is shifted to higher energies.

**XUV Transient Absorption Measurements.** The time evolution of changes to the  $M_{2,3}$ -edge absorption spectra are shown in the FXTAS spectrograms in Fig. 3 *A* and *B*, for the metallic and insulating phases, respectively. In both cases, the system displays an abrupt response due to the initial photoexcitation and reaches a steady state within less than 100 fs, with the insulating phase equilibrating significantly faster than the metallic phase. The observed changes persist out to >60 ps (*Picosecond XUV Transient Absorption Measurements*). To compare both data sets the  $\Delta\text{OD}$  changes must be plotted with respect to the same reference spectrum. The metallic phase is treated as the sum of  $\Delta\text{OD}$  changes first upon heating and subsequently upon photoexcitation (i.e., as the sum of the curves in Fig. 3*C*, *Upper*) yielding the purple curve in the lower panel). This gives the cumulative  $\Delta\text{OD}$  changes with respect to the insulating-phase static spectrum, which can be compared with the signal at late time delays measured in the insulating phase directly (shown as the orange curve in Fig. 3*C* (*Lower*)). The two core-level spectra in the lower panel match very closely, implying the final state in each case is the same equilibrated long-lived excited state of the metallic phase. Measurements of the fluence dependence (*Fluence Dependence Measurements*) further confirm that the signal observed in the insulating phase displays a critical fluence threshold

and saturation, confirming that it indeed corresponds to the phase transition.

**Analysis of the Metallic Phase.** The metallic-phase dynamics highlight the significant effects that carrier excitation can have on the spectrum, independent of the phase transition. The initial excitation with an NIR pulse excites dominantly ligand-to-metal charge-transfer-type transitions, which affect the allowed XUV dipole transitions and alter the shape of the spectrum. The dynamics can be globally fit with a two-component model, with an early-time spectral component corresponding to the nonequilibrium charge distribution immediately following photoexcitation and a late-time spectral component corresponding to the charge distribution following carrier-carrier and intervalley (carrier-phonon) scattering. The global fit is performed using first-order kinetics (*Global Fitting*) and the calculated spectral components and their corresponding kinetic behaviors are illustrated in Fig. 4 *A* and *B*, respectively. The tensor product of these curves gives the full 2D global fit that is shown in Fig. 3*D*. The global fit captures all of the observed dynamics and is a more rigorous way to account for broadband effects, as opposed to fitting of single lineouts. A  $69 \pm 11$ -fs time constant is obtained for the metallic phase kinetics (orange and blue curves, Fig. 4*B*).



**Fig. 3.** Transient absorption spectrograms are shown for the metallic phase (*A*) and insulating phase (*B*) of  $\text{VO}_2$ , at an incident fluence of  $25 \text{ mJ/cm}^2$ . (*C*) Comparison of the systems at late-time delays shows that the insulator and metal reach the same final state. The metallic-phase  $\Delta\text{OD}$  with respect to the insulating-phase static spectrum, shown in purple (*Lower*), is equal to the sum of the changes first upon heating (red curve, *Upper*) followed by photoexcitation (blue curve, *Upper*). Note that the red curve is identical to the difference spectrum in Fig. 2*A*. The global fits for the metallic and insulating phases are shown in *D* and *E*, respectively.



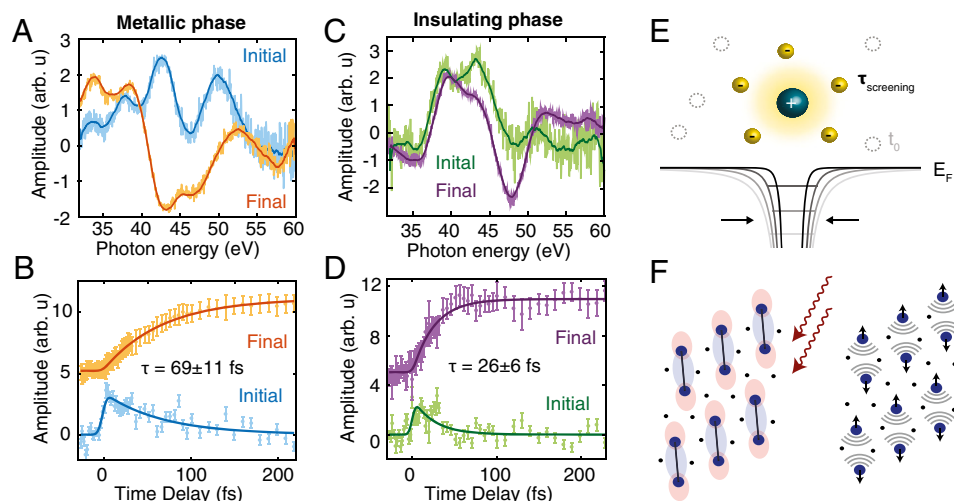
**Analysis of the Insulating Phase.** In the insulating phase, as in the metallic phase, the time-dependent data are well described by single-exponential kinetics. Biexponential kinetics, related to a separable observation of carrier thermalization and the IMT, are not discernible in the data. The insulating phase is thus also fit by a two-component model, which describes the evolution of an initial excitation to a final charge distribution, encompassing the IMT. The phase transition proceeds directly into an excited state of the metallic phase, as was found by the comparison at late-time delays with the photo-excited metallic phase, shown in Fig. 3C (*Lower*). The spectral signatures of the initial and final states of the insulator are shown in Fig. 4C, and the corresponding kinetics are shown in Fig. 4D. The time constants of both the rise and decay (Fig. 4D, purple and green curves) are globally fit with a value of  $26 \pm 6$  fs. The formation of this final state represents an IMT into a long-lived excited state of the metallic phase, which persists out to more than +60 ps (*Picosecond XUV Transient Absorption Measurements*). Fluence-dependent measurements (*Fluence Dependence Measurements*) further confirm that this signal displays the characteristic threshold fluence threshold and saturation behavior associated with the phase transition.

As noted, the phase transition and carrier thermalization processes are not distinguishable in the data of the insulating phase. As described in *Global Fitting*, a three-component fit that attempts to separate these effects retrieves only one timescale for both the carrier thermalization and the insulator-to-metal transition, and no signature of the longer timescale measured in the metallic phase is discernible. This may be due to differences in the excitation step, which in the metallic phase exhibits more ligand-to-metal charge-transfer-type transitions (O 2p to V 3d bands) and the insulating phase exhibits more  $\sigma \rightarrow \sigma^*$  (or  $d_{||}$  to  $d_{||}^*$ )-type transitions, as illustrated in Fig. 1. In the final state, electrons and holes thermalize to the band edge, which has d-band character; thus this process may be apparent only in the metallic case, where the scattering processes result in a net change of oxidation state.

## Discussion

The XUV measurements directly probe the electronic response of the vanadium atoms in the bulk material, establishing that the ultrafast response is not limited to the surface (23) or to single-crystal samples (25). The  $26 \pm 6$ -fs timescale observed is notably slower than the inverse plasma frequency, often associated with screening, which is calculated to be about 4 fs for the excitation density in these experiments (*Carrier Density and Plasma Frequency Estimate*). However, there is some evidence that purely electronic bottlenecks could be longer for Mott transitions (40). The electronic screening time may be a complex function of orbital-specific carrier occupation and mobility, where only a subset of the excited carriers can participate in the relevant local screening. A free-carrier screening-mediated Mott transition is schematically illustrated in Fig. 4E. Upon exciting an electron-hole plasma, the carriers are spatially distributed according to the oscillator strengths and joint density of states of the interband transitions. As Coulomb forces drive these carriers to new equilibrium spatial distributions the ionic vanadium sites of the lattice are progressively screened. The screened Coulomb potential (frequently described by a Yukawa potential, as illustrated in Fig. 4E) narrows and can reach a critical point where the d electrons, initially localized on the vanadium sites, are no longer bound and the system becomes metallic, as in a doping-driven Mott transition (41–43).

Considering a structural involvement, the  $26 \pm 6$ -fs timescale observed for the IMT is well below the 160-fs period of the  $A_g$  phonon mode previously thought to create a structural bottleneck for the phase transition (22). The fastest phonons previously hypothesized to play a role in the IMT have a period of 98 fs (44) and are related to the V–V dimer stretch. However, the timescale measured here is more consistent with the half period of octahedral V–O–V bending modes at  $620 \text{ cm}^{-1}$ , which have a period of 55 fs (45). New lattice vibrations could also occur on the excited state potential, which is altered as a result of pump-pulse-induced changes in orbital occupation (24, 28, 46). The process is schematically illustrated in Fig. 4F. The resulting displacement can alter the overlap of atomic orbitals and screening,



**Fig. 4.** (A) Spectral signatures of the initial (blue) and final (orange) charge distributions observed in the metallic-phase data. Statistical error is apparent from the line fluctuations. The kinetic behavior of each of these spectra in the metallic phase is shown in B, where corresponding spectra and kinetics are shown in identical colors. The data points are the result of fitting the  $\Delta OD$  spectrum at each time delay to a superposition of the two spectral curves in A. The results highlight the quality of agreement of the global fit. Analogous spectral and kinetic fit components for the insulating phase are plotted in C and D, respectively, showing much faster kinetics. E illustrates how the phase transition can occur in time as the carriers redistribute spatially to screen the ion cores. The potential well changes from a Coulomb potential (light gray) to a screened Coulomb, or Yukawa potential (dark gray to black), while the horizontal lines represent the binding energy of the d electron, leading up to the Fermi level. F illustrates the lattice response, where a change in the excited-state potential energy surface (red: increase, blue: decrease) may trigger impulsive nuclear motion via phonon excitation.

leading to a bandwidth-driven Mott transition (43). While a 1D Peierls-type mechanism has often been discussed, the classical Peierls instability is a feature of  $d^1$  systems (47), whereas  $\text{VO}_2$  is determined here to be more accurately  $d^2$ . Additionally, the short IMT timescale measured here, which does not align well with vibrational modes typically associated with a Peierls transition (22, 44), lends support to a Mott-like picture.

The timescale measured in this work is consistent with recent femtosecond photoemission (23) and transient reflectivity (25) results on the IMT in  $\text{VO}_2$  that yielded 60- and 40-fs (pulse-duration-limited) timescales, respectively, but is not limited by the instrument response function of the measurement. The measurement of a non-pulse-duration-limited  $26 \pm 6$ -fs timescale supports a Mott-like mechanism as opposed to Peierls, but does not rule out structural involvement. Electronic and structural effects may indeed be working cooperatively (28). A full theoretical description of the dynamics would likely need to incorporate the electron hopping time, or transfer integral, as well as screening to fully capture the role of both energy shifts and bandwidth changes across the phase transition. The observation of a common nonthermal final state is similar to recent ultrafast pump-probe measurements on the insulating and metallic phases of the prototypical Mott-Hubbard transition in  $\text{V}_2\text{O}_3$  (48). Further elucidation will require more advanced theory, mapping screening and structural distortion effects onto the XUV spectrum—an exciting and tangible possibility.

The time constant measured here is not necessarily a fundamental lower limit on the timescale of the IMT process in  $\text{VO}_2$ . It may vary and indeed be faster based on sample crystallinity, doping, strain, and starting temperature (25, 49), as well as on the pulse spectrum and energy of the pump beam, which determines the initial spatial and energy distribution of the injected charges. While all of the previous ultrafast experiments on  $\text{VO}_2$  (as well as this work) have used Ti:sapphire laser pulses, the bandwidth and central wavelength of the pulses are varied—however, as most of these results were ultimately limited by the temporal resolution of the experiment, it is not yet possible to map out a trend as to how these factors influence the dynamics. In regard to sample crystallinity, results to date are generally in agreement between single-crystal and polycrystalline films: time-resolved electron diffraction has measured structural timescales in agreement (26, 27), and the fastest electronic timescales measured for each case were previously limited by the pulse duration [ $<60$  fs for thin films (23), and  $<40$  fs for microrods (25)]. We may speculate that excitation closer to the band edge and in single-crystal samples would provide the “natural” speed limit; however, based on this work, dynamics in polycrystalline samples may actually be faster, and more data will be needed to understand this. Broadband XUV attosecond pulses provide access to core-level transitions with temporal resolution on the order of electronic timescales, making FXTAS a powerful tool to study these effects. For  $\text{VO}_2$ , it is shown that the IMT occurs within  $26 \pm 6$  fs following a change in orbital occupancy. Together with global analysis and theoretical modeling, this methodology has the potential to disentangle the effects of spin, lattice, and orbital degrees of freedom in other correlation-driven phenomena in condensed matter systems.

## Materials and Methods

**Samples.** Experiments were carried out on 25-nm-thick polycrystalline  $\text{VO}_2$  thin films prepared by pulsed laser deposition (50) in an Epion PLD-3000 system (base vacuum of  $9 \times 10^{-7}$  torr). A vanadium metal target was ablated at room temperature in an oxygen atmosphere ( $1.1 \times 10^{-2}$  torr maintained by 2-cm flow) by an excimer laser (248 nm, Lambda Physik Comp-Ex 205, nominal pulse duration 25 ns), at a fluence of  $4 \text{ J/cm}^2$  with a pulse repetition rate of 25 Hz. The average deposition rate of the  $\text{VO}_x$  film was  $0.3 \text{ \AA/s}$ . After deposition, the films were annealed in a tube furnace at  $450^\circ\text{C}$  for 10 min in 250 mtorr oxygen to produce stoichiometric  $\text{VO}_2$ . The samples were deposited on 30-nm-thick freestanding  $\text{Si}_3\text{N}_4$  membranes provided by Silson Ltd. The clear membrane and sample area was  $3 \text{ mm} \times 3 \text{ mm}$ , and the membrane was supported on a silicon frame of dimensions  $7.5 \text{ mm} \times 7.5 \text{ mm}$ .

**XUV Measurements.** FXTAS was performed in a pump-probe scheme using few-femtosecond carrier-envelope phase-stabilized NIR pump pulses and attosecond probe pulses. Pulses from a Femtopower Compact Pro amplifier were compressed to 5-fs duration and 750- $\mu\text{J}$  pulse energy using self-phase modulation in a neon-filled (2-bar) hollow-core fiber and a chirped mirror compressor (optics from Ultrafast Innovations). The compressed pulses had a continuous spectrum spanning from 400 to 1,000 nm and were characterized using dispersion scan [Sphere Photonics (51)] to have a duration of 4.9 fs. The 1-kHz repetition rate was reduced to 100 Hz using an optical chopper to avoid sample damage, and the beam was split into two arms using an 80–20% broadband beam splitter.

Probe pulses were generated using the 80% majority arm, by focusing  $\sim 400\text{-}\mu\text{J}$  5-fs pulses into a krypton gas target to generate a sub-5-fs attosecond pulse train with a continuous spectrum of XUV light spanning from 30 to 60 eV via high harmonic generation. Residual NIR light from the generation process was removed by transmitting the XUV through a 200-nm-thick aluminum filter. The XUV pulse was focused through the  $\text{VO}_2$  sample using a grazing-incidence gold-coated toroidal mirror (ARW Corporation). The 20% pump arm was time-delayed using a retro-reflector on a piezo delay stage with subnanometer positional stability (Physik Instrumente). The pump beam was then focused through the sample collinearly with the probe beam by recombining on a  $45^\circ$  annular mirror, where the pump arm is reflected and the probe arm is transmitted. The NIR beam was focused to  $\sim 200\text{-}\mu\text{m}$  FWHM on the sample, and the XUV beam to  $\sim 60\text{-}\mu\text{m}$  FWHM. After the sample the NIR beam was blocked using a pinhole and a 200-nm aluminum filter. The XUV transmission spectrum through the sample was diffracted using a grazing-incidence variable-line-spacing reflection grating (Hitachi) and recorded on an XUV CCD camera (Princeton Instruments PIXIS). The spectral resolution was determined to be  $\sim 50 \text{ meV}$  at 45 eV, calibrated using the linewidths of the  $2s^21p^6np$  ( $n = 3, 4, 5, \dots$ ) Rydberg series of inner-valence excitations in neon (52).

Static absorption spectra were recorded using only the XUV beam and alternating between a sample and a reference target, i.e., a  $\text{VO}_2$  sample and a bare  $\text{Si}_3\text{N}_4$  membrane. The samples were alternated rapidly using fast vacuum translation stages (Physik Instrumente) to compensate for fluctuations and systematic drifts of the high harmonic flux and obtain spectra with sub-1-m $\Delta\text{OD}$  accuracy. Spectra of the metallic phase of  $\text{VO}_2$  were obtained by heating the sample in vacuum using a quartz halogen lamp. Samples were checked for uniformity both via optical microscope and by performing a transmission map measurement in the XUV, where the integrated XUV flux was measured as a function of 2D position on the sample surface, and found not to vary by more than  $\pm 5\%$ .

Transient absorption spectra were recorded using pump-on/pump-off measurements, where at each time delay a spectrum is saved first with the pump beam switched off and then with the pump beam switched on by means of a computer-controlled optical shutter. This procedure allows spectra to be recorded rapidly back-to-back for higher-quality data. The transient absorption measurements were taken in 2-fs steps from  $-30$  to  $+30$  fs, in 5-fs steps from  $+30$  to  $+100$  fs, and in 10-fs steps from 100 to 220 fs, with a repetition rate of 100 Hz and an exposure time of 1 s. To prevent accumulation of heat in the sample it is critical that subsequent laser pulses do not impinge on the same area of the sample, as heat dissipation in the thin-film samples in vacuum is comparable to the 10-ms inverse laser repetition rate after the optical chopper. To counteract this, the samples were rapidly rotated normal to their surface by mounting the sample in a hollow-shaft motor rotating at  $37 \pm 3 \text{ Hz}$ , and the beam was transmitted at a 0.75-mm radius relative to the motor axis. The material's response in the metallic phase was measured by performing the transient absorption measurement with the heating lamp on as a reference. It was confirmed that without the rotating motor turned on, and using a slower raster-scanning method, where multiple subsequent pump pulses at 100 Hz can impinge on the same spot on the sample before moving to a new area, the samples displayed the dynamic response of the metallic phase. With the spinning motor turned on, the response of the insulating phase could be clearly isolated, evidenced by the absence of any significant signal at negative delays and a qualitatively different measured transient signal. A transient absorption experiment was also performed on a bare  $\text{Si}_3\text{N}_4$  substrate and it was confirmed that the substrate material was optically transparent and inert to the pump pulses.

Further details can be found in [Supporting Information](#). Data are available on request to the corresponding author.

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